

Cite this: *Chem. Commun.*, 2012, **48**, 11674–11676

www.rsc.org/chemcomm

COMMUNICATION

A stable sulfone based electrolyte for high performance rechargeable Li–O₂ batteries†Dan Xu,^a Zhong-li Wang,^a Ji-jing Xu,^a Lei-lei Zhang,^{ab} Li-min Wang^a and Xin-bo Zhang^{*a}

Received 19th September 2012, Accepted 12th October 2012

DOI: 10.1039/c2cc36815c

A tetramethylene sulfone based electrolyte was used in rechargeable Li–O₂ batteries for the first time. Without a catalyst, superior battery performances were successfully obtained, and desired lithium peroxide (Li₂O₂) dominates the discharge product.

Rechargeable lithium–O₂ batteries are the subject of significant interest because of their extremely high energy density.¹ A typical rechargeable non-aqueous Li–O₂ cell is composed of a Li metal anode, a non-aqueous Li⁺ conducting electrolyte, and a porous cathode. During discharge, oxygen is first reduced to superoxide (O₂[−]) and supposed to react with Li⁺ to form unstable intermediate lithium superoxide (LiO₂), followed by disproportion into the principal reversible discharge product, lithium peroxide (Li₂O₂) (overall reaction: 2Li⁺ + O₂ + 2e[−] ↔ Li₂O₂). Until now, although much progress has been made, Li–O₂ batteries are still in their infancy and great improvements are urgently needed, with the top of the list being the cathode and especially the electrolyte.²

An ideal electrolyte for non-aqueous Li–O₂ batteries should possess many important physicochemical properties, including low volatility, high oxygen solubility, and in particular, to ensure repeated and highly reversible formation and decomposition of Li₂O₂ on cycling. However, none of the reported electrolytes can completely meet these requirements.^{3a} For example, early investigations of non-aqueous Li–O₂ cells focused on the use of organic carbonate based electrolytes due to their high boiling point and ion conductivity. Unfortunately, Li–O₂ batteries with carbonate based electrolytes suffer from insufficient capacity, high charge potential, low energy efficiency, and irreversible decomposition at the cathode on discharge to form unwanted products such as lithium carbonate (Li₂CO₃) and other organic lithium salts with little or none of the desired product (Li₂O₂).^{3b–e} Later research turned to studying ethers such as dimethoxy ethane (DME), 1,3-dioxolane, and tetraethylene glycol dimethyl ether (TEGDME) as electrolytes.

Although Li₂O₂ was shown to be dominant product, especially with a TEGDME based electrolyte, electrolyte decomposition on cycling still could be not absolutely excluded.⁴ In addition, ether electrolytes are often plagued by high volatility (DME) and/or high viscosity (TEGDME), which might limit the cycle life and/or rate performance of Li–O₂ batteries. Up to now, although the detailed mechanism of electrolyte decomposition during the processes that occur on discharge and subsequent charge is still not absolutely clear, it is beyond doubt that the very active intermediates (O₂[−], LiO₂, etc.) should be blamed most for the poor reversibility and cycle life of Li–O₂ batteries. Therefore, development of novel electrolytes by investigating their behaviour in an O₂/O₂[−] environment and further studying the electrochemical products and reversibility in Li–O₂ batteries is urgently important.

To this end, we have recently studied the electrochemical performance of dimethyl sulfoxide (DMSO) based electrolyte in Li–O₂ batteries. Interestingly, with this novel electrolyte and a common carbon cathode, high capacity and good reversibility were obtained.^{5a} Very recently, Bruce *et al.* further improved the performance of Li–O₂ batteries with DMSO by using porous Au as the cathode.^{5b} Although DMSO has been shown to be a powerful solvent for Li–O₂ batteries, it is still unsatisfactory when considering that the formation of Li₂O₂ is still accompanied by unwanted by-products when carbon is employed as the cathode material.^{5a} This led us to further consider why DMSO can stabilize O₂[−] but not enough for Li₂O₂, and whether there are other solvents that inherit the advantages of DMSO and also support more robust Li₂O₂ formation.

Tetramethylene sulfone (TMS), as a universal dipolar solvent, is a kind of sulfone which has the oxidation state of sulfoxide. Besides high solubility, low toxicity, superior safety as well as sufficient ion dynamics,⁶ TMS also has superior properties to DMSO, including much lower volatility and higher resistance to electrochemical oxidation (Table S1, ESI†).⁷ Therefore, TMS possesses many desirable properties but its electrochemical behaviour in a Li⁺ containing environment and its performance in Li–O₂ batteries remains unexplored.

Herein, we employed TMS as a novel electrolyte solvent and evaluated its performance in Li–O₂ batteries. Interestingly, superior performances including high capacity, good reversibility and rate performance were achieved, and the performance could be improved by using an optimized graphene

^a State Key Laboratory of Rare Earth Resource Utilization, Changchun Institute of Applied Chemistry, Chinese Academy of Sciences, Changchun, 130022, Jilin, China. E-mail: xbzhang@ciac.jl.cn; Fax: +86 431 85262235; Tel: +86 431 85262235

^b University of Chinese Academy of Sciences, Beijing, 100049, China
† Electronic supplementary information (ESI) available: Experimental details and supplementary figures. See DOI: 10.1039/c2cc36815c

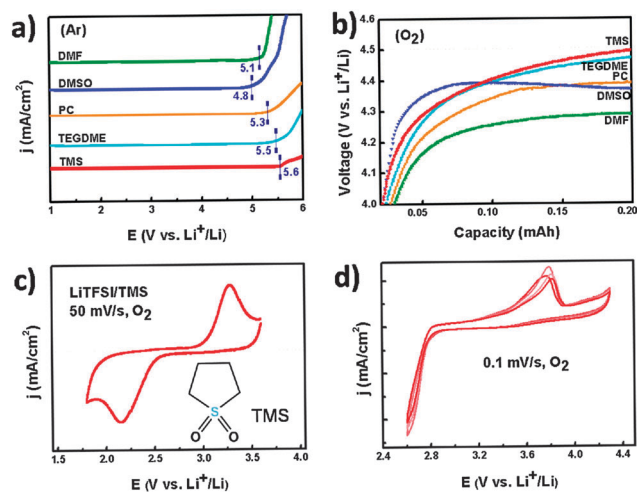


Fig. 1 (a) Electrochemical windows of solvents commonly used in Li–O₂ batteries. (b) Decomposition voltage of solvents in a Li–O₂ battery under an O₂ atmosphere. (c) CV curve of 1 M LiTFSI/TMS electrolyte saturated with O₂. (d) CV curves of a Li–O₂ battery with a TMS based electrolyte.

oxide derived carbon cathode. Furthermore, Li₂O₂ was confirmed to be the dominant discharge product.

The basic electrochemical properties of common electrolytes used in Li–O₂ batteries were firstly investigated. As clearly shown in Fig. 1a, under an Ar atmosphere, the oxidation potentials were found to be 5.3, 5.5, 5.1, 4.8, and 5.6 V *versus* Li⁺/Li for PC, TEGDME, DMF, DMSO, and TMS, respectively, indicating that their electrochemical stability increases following the order of DMSO < DMF < PC < TEGDME < TMS. This demonstrates that TMS possesses the highest resistance toward electrooxidation. It is well known that an electrolyte for a Li–O₂ battery should endure high overpotential during charge and thus superior antioxidant ability is a prerequisite to obtain a stable Li–O₂ battery. Therefore, based on the above results we can reasonably expect that a Li–O₂ battery with TMS could have the best stability under an oxygen atmosphere. To confirm this point, the decomposition voltages of the solvents were examined in a Li–O₂ battery during charge without the previous discharge process. As shown in Fig. 1b, the stability decreases with a sequence similar to that of the electrochemical window (TMS > TEGDME > PC > DMSO > DMF). TMS shows the best stability (no decomposition until 4.5 V), illustrating the favourability of TMS as a stable solvent for Li–O₂ batteries.

The reversibility of electrochemical reactions in the TMS based electrolyte saturated with O₂ was then investigated. Only one pair of well-defined reduction/oxidation peaks could be found (Fig. 1c), which can be attributed to the reversible formation/decomposition of Li₂O₂. This result is consistent with the CV curves of Li–O₂ cells with a TMS based electrolyte (Fig. 1d).

Fig. 2a shows the first discharge and charge curves of Li–O₂ batteries with a KB carbon based cathode. It was found that the discharge voltage plateau is 2.7 V and the specific capacity is 7735 and 6305 mAh g⁻¹, for the discharge and charge processes at a current density of 0.1 mA cm⁻² respectively, corresponding to a high round-trip efficiency of 81.5%.

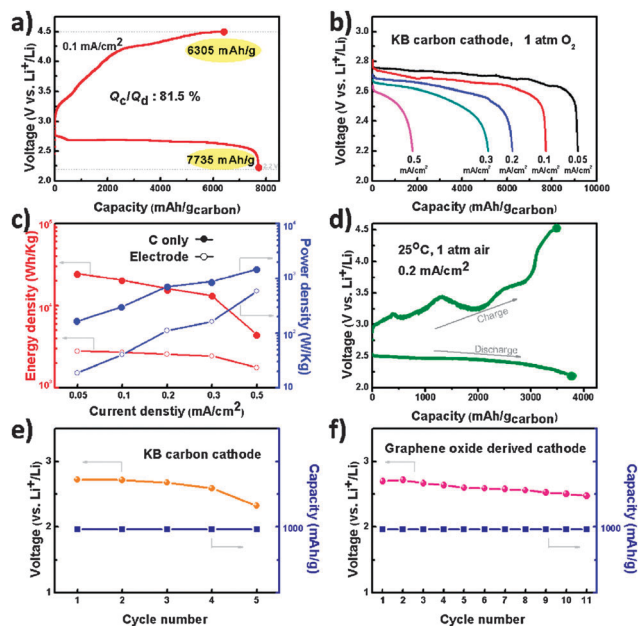


Fig. 2 Li–O₂ battery performance at room temperature under 1 atm O₂ atmosphere. (a) First discharge/charge curve at a current density of 0.1 mA cm⁻², (b) rate capability, and (c) gravimetric energy and power density. (d) Discharge/charge curve of a Li–air battery under a current density of 0.2 mA cm⁻². (e) Cycle stability of Li–O₂ batteries with a KB carbon cathode and (f) with an optimized cathode.

The rate capacity demonstrated in Fig. 2b indicates that an even higher capacity of 9100 mAh g⁻¹ is obtained at a lower current density of 0.05 mA cm⁻². Although the discharge capacity decreases with increasing current density, the Li–O₂ battery with TMS still shows a high capacity of 1700 mAh g⁻¹ at a high current density of 0.5 mA cm⁻². Fig. 2c shows the energy and power density of a Li–O₂ battery with a TMS based electrolyte and a KB carbon cathode. Interestingly, it was found that the energy density reaches up to 24679 W h kg⁻¹ by carbon weight and 2790 W h kg⁻¹ by whole weight of the electrode (C + PVDF + Li₂O₂). The electrochemical performances obtained are much higher than those of the most advanced Li ion battery,^{8a} and are quite impressive for a Li–O₂ battery with a carbon based cathode (without catalysts).^{3b,5b}

The electrochemical performance of the TMS based electrolyte in a Li–air battery was also investigated (Fig. 2d), and although the O₂ partial pressure decreases significantly in air, a high capacity of 3700 mAh g⁻¹ was still obtained. It should be noted that compared to a Li–O₂ battery, the charge profile of the Li–air battery is quite complicated, which might be due to complex reactions caused by CO₂ and H₂O in the air.

The cycle performance was evaluated by controlling discharge depth to 1000 mAh g⁻¹. As shown in Fig. 2e and S2 (ESI[†]), a Li–O₂ battery with a TMS based electrolyte can keep 5 cycles on a conventional KB carbon cathode, which is similar to those observed for TEGDME and DMSO based electrolytes using the same cell technology. It should be noted that after substituting the KB carbon cathode with an advanced graphene oxide derived cathode,^{8b} cycle performance improved significantly (Fig. 2f and S3, ESI[†]), which might indicate that the poor quality of the O₂ electrode,

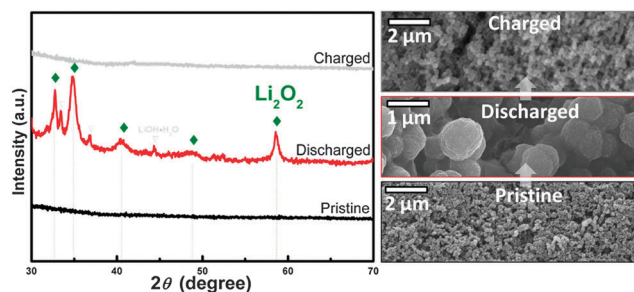


Fig. 3 XRD profiles and SEM images of pristine, discharged, and recharged O₂ electrode of a Li–O₂ battery with a TMS based electrolyte.

and not the solvent itself, should be mainly responsible for the limited cycle life of the Li–O₂ battery. These facts further support the feasibility of TMS as a solvent for Li–O₂ batteries, and we believe that the performance can be further improved by employing suitable combinations of electrolyte, cathode, and cell technology.^{5b,8c}

The morphology of the O₂ electrode before and after discharge was then investigated using scanning electron microscopy (SEM). As shown in Fig. 3, after a deep discharge, the porous O₂ electrode is fully covered by a large number of spheres with a wide diameter dispersion (100 nm ~ 1 μm). The powder X-ray diffraction (XRD) profile illustrates that the dominant product is well crystallized Li₂O₂. Meanwhile, a negligible amount of lithium hydroxide hydrate could also be detected as a by-product. After recharge, all of the crystallization peaks in the XRD profile disappear, indicating that the discharge products decompose thoroughly, which is consistent with the SEM image. Both SEM and XRD analysis indicate the good reversibility of Li–O₂ batteries with a TMS based electrolyte.

Now it is well considered that the formation and dominance of Li₂O₂ in the discharge product is critical to the operation of a Li–O₂ battery. Interestingly, the dominance of Li₂O₂ can easily be found in Li–O₂ batteries with a TMS based electrolyte (Fig. S4a and S4b, ESI†). On the other hand, besides Li₂O₂, a large amount of nanosheet-like LiOH is observed in Li–O₂ batteries with DMSO (Fig. S4c and S4d, ESI†), implying that there are fewer possible side reactions in a Li–O₂ battery with TMS compared to one with DMSO. The underlying mechanism might be related to the kinetics of superoxide and/or Li₂O₂-induced decomposition of solvent, PVDF, or C in the Li–O₂ battery, which can be greatly affected or controlled by the solvent (Fig. S5, ESI†).

Here, we firstly propose and demonstrate a novel TMS based electrolyte with attractive physicochemical properties for rechargeable Li–O₂ batteries. Interestingly, superior performances including high capacity, good round-trip efficiency and cycle life especially when combined with an optimized O₂ electrode were successfully obtained. The dominant discharge product in this novel electrolyte was confirmed to be the

desired Li₂O₂. The promising results obtained are believed to open new and exciting possibilities to promote the development of rechargeable Li–O₂ batteries.

This work is financially supported by the 100 Talents Programme of the Chinese Academy of Sciences, the National Program on Key Basic Research Project of China (973 Program, Grant No. 2012CB215500), the Foundation for Innovative Research Groups of the National Natural Science Foundation of China (Grant No. 20921002), and the National Natural Science Foundation of China (Grant No. 21101147, 21271168, and 21203176).

Notes and references

- (a) K. M. Abraham and Z. Jiang, *J. Electrochem. Soc.*, 1996, **143**, 1; (b) J. Read, *J. Electrochem. Soc.*, 2002, **149**, A119; (c) T. Ogasawara, A. Debart and P. G. Bruce, *J. Am. Chem. Soc.*, 2006, **128**, 1390; (d) Y. C. Lu, Z. C. Xu and S. H. Yang, *J. Am. Chem. Soc.*, 2010, **132**, 12170; (e) Y. F. Wang, D. Zheng and D. Y. Qu, *Energy Environ. Sci.*, 2011, **4**, 3697; (f) E. Yoo and H. S. Zhou, *ACS Nano*, 2011, **5**, 3020; (g) Y. L. Li, J. J. Wang and X. L. Sun, *Chem. Commun.*, 2011, **47**, 9438; (h) Y. M. Cui, Z. Y. Wen and Y. Liu, *Energy Environ. Sci.*, 2011, **4**, 4727; (i) T. Kuboki, T. Okuyama and N. Takami, *J. Power Sources*, 2005, **146**, 766; (j) J. Xiao, J. Hu and Z. J. Nie, *J. Power Sources*, 2011, **196**, 5674; (k) S. M. Dong, X. Chen and G. L. Cui, *Chem. Commun.*, 2011, **47**, 11291; (l) L. L. Zhang, X. B. Zhang, Z. L. Wang, J. J. Xu, D. Xu and L. M. Wang, *Chem. Commun.*, 2012, **48**, 7598.
- (a) P. G. Bruce, S. A. Freunberger and L. J. Hardwick, *Nat. Mater.*, 2012, **11**, 19; (b) J. Christensen, P. Albertus and A. Kojic, *J. Electrochem. Soc.*, 2012, **159**, R1; (c) M. G. Park, H. Y. Sun and J. Cho, *Adv. Energy Mater.*, 2012, **2**, 780; (d) R. Black, B. Adams and L. F. Nazar, *Adv. Energy Mater.*, 2012, **2**, 801; (e) G. Girishkumar, B. McCloskey and A. C. Luntz, *J. Phys. Chem. Lett.*, 2010, **1**, 2193.
- (a) D. T. Sawyer and J. S. Valentine, *Acc. Chem. Res.*, 1981, **14**, 393; (b) S. A. Freunberger, Y. Chen, Z. Peng, J. M. Griffin, L. J. Hardwick, F. Bardé, P. Novák and P. G. Bruce, *J. Am. Chem. Soc.*, 2011, **133**, 8040; (c) W. Xu and V. V. Viswanathan, *J. Power Sources*, 2011, **196**, 3894; (d) V. S. Bryantsev, M. Blanco and F. Faglioni, *J. Phys. Chem. A*, 2011, **115**, 12399; (e) V. S. Bryantsev and F. Faglioni, *J. Phys. Chem. A*, 2012, **116**, 7128.
- (a) T. Laino and A. Curioni, *Chem.–Eur. J.*, 2012, **18**, 3510; (b) C. O. Laoire, S. Mukerjee and K. M. Abraham, *J. Electrochem. Soc.*, 2011, **158**, A302; (c) S. A. Freunberger, Y. Chen, N. E. Drewett, L. J. Hardwick, F. Bardé and P. G. Bruce, *Angew. Chem., Int. Ed.*, 2011, **50**, 8609; (d) J. Hassoun, F. Croce and B. Scrosati, *Angew. Chem., Int. Ed.*, 2011, **50**, 2999.
- (a) D. Xu, Z. L. Wang and X. B. Zhang, *Chem. Commun.*, 2012, **48**, 6948; (b) Z. Q. Peng, S. A. Freunberger and P. G. Bruce, *Science*, 2012, **337**, 563.
- (a) A. Abouimrane, I. Belharouak and K. Amine, *Electrochem. Commun.*, 2009, **11**, 1073; (b) X. G. Sun and C. A. Angell, *Electrochem. Commun.*, 2009, **11**, 1418; (c) A. Lewandowski and A. Swiderska-Mocek, *Electrochim. Acta*, 2011, **56**, 5972.
- (a) T. L. Buxton and J. A. Caruso, *J. Phys. Chem.*, 1973, **77**, 1882; (b) Y. Matsuda, M. Morita and K. Hirai, *J. Electrochem. Soc.*, 1985, **132**, 2538; (c) U. Tilstam, *Org. Process Res. Dev.*, 2012, **16**, 1273; (d) N. Sahoo, X. G. Sun and D. E. Jiang, *J. Phys. Chem. B*, 2012, **116**, 3235.
- (a) R. M. Robert, M. G. Betar and S. H. Yang, *Energy Environ. Sci.*, 2011, **4**, 2952; (b) Z. L. Wang, D. Xu and X. B. Zhang, *Adv. Funct. Mater.*, 2012, **22**, 3699; (c) H. G. Jung, J. H. Hassoun and B. Scrosati, *Nat. Chem.*, 2012, **4**, 579.